# Long-term antibacterial properties of a nanostructured titanium alloy surface: An *in vitro* study

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## ARTICLE INFO

## ABSTRACT

The demand for joint replacement and other orthopedic surgeries involving titanium implants is continuously increasing; however, 1%–2% of surgeries result in costly and devastating implant associated infections (IAIs). *Pseudomonas aeruginosa* and *Staphylococcus aureus* are two common pathogens known to colonise implants, leading to serious complications. Bioinspired surfaces with spike-like nanotopography have previously been shown to kill bacteria upon contact; however, the longer-term potential of such surfaces to prevent or delay biofilm formation is unclear. Hence, we monitored biofilm formation on control and nanostructured titanium disc surfaces over 21 days following inoculation with *Pseudomonas aeruginosa* and *Staphylococcus aureus*. We found a consistent 2-log or higher reduction in live bacteria throughout the time course for both bacteria. The biovolume on nanostructured discs was also significantly lower than control discs at all time points for both bacteria. Analysis of the biovolume revealed that for the nanostructured surface, bacteria was killed not just on the surface, but at locations above the surface. Interestingly, pockets of bacterial regrowth on top of the biomass occurred in both bacterial species, however this was more pronounced for *S. aureus* cultures after 21 days. We found that the nanostructured surface showed antibacterial properties throughout this longitudinal study. To our knowledge this is the first *in vitro* study to show reduction in the viability of bacterial colonisation on a nanostructured surface over a clinically relevant time frame, providing potential to reduce the likelihood of implant associated infections.

## 1. Introduction

Orthopedic implants and other devices play a crucial role in restoring health, treating musculoskeletal disease, and ultimately saving patient lives [1]. Titanium alloy is widely used as a biomaterial to replace hard tissue, due to its mechanical bone-like properties, enhanced corrosion resistance and superior biocompatibility [2]. However, implant associated infection (IAI) occurs in 1%–2% of joint replacement [3,4] and approximately 30% of open fracture reduction cases [5,6], leading to notably increased morbidity, mortality, and treatment cost [7,8]. IAI involves a series of complex interactions between the pathogen, the implant, and the host immune response. In the absence of a foreign body, tissue contamination by opportunistic pathogens is usually spontaneously cleared by host immune defences. However, the presence of an implant triggers a localised tissue response, which includes acute and chronic inflammation, a foreign body reaction, formation of granulation

tissue and, finally, fibrous encapsulation. In this early phase, bacteria may take advantage of the immunocompromised region around the implant to adhere, colonise and form a biofilm [9]. Ideally, in a concept termed 'race to the surface', host cells outcompete bacteria to colonise the implant surface, leading to favourable tissue integration – where the host immune system is better placed to prevent the onset of IAI [10,11].

Bacteria adhere to implants by a process mediated by hydrophobic, electrostatic and van der Waals interactions, where they colonise the device surface to form a biofilm which can evade the immune response. A biofilm manifests, following a sequence of growth phases to become a multifaceted biological unit that is extremely difficult to treat, becoming resistant to systemic antibiotics [12–14]. An IAI results in an unresolved chronic inflammatory response, leading to tissue destruction and ultimately loosening of the bone-implant interface [15].

Treatment of an IAI usually involves the removal of the implant and associated cement, debridement of all devitalised tissue, and long-term

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antimicrobial treatment [16]. Sources of bacterial contamination implicated in IAI include the operating room environment, surgical equipment, clothing worn by medical staff, and bacteria on the patient's skin and residing in the patient's body [17–19]. The most common etiologic agents causing IAIs are *Staphylococcus aureus* (*S. aureus*), *Staphylococcus epidermidis (S. epidermidis)* [20], *Pseudomonas aeruginosa* (*P. aeruginosa*) [21], and *Escherichia coli* (*E. coli*) [22]. These bacteria are normally associated with skin and environmental microbiota but can quickly colonise implants leading to biofilm formation [23]. Four out of five IAIs are caused by *Staphylococcus* species, with *S. aureus* and *S. epidermidis* responsible for two out of three [20]. Drug-eluting coatings have been successfully attached to implants, including silver nanoparticles [24–26], antibacterial peptides [27,28] and antibiotics [29], amongst others [30, 31]. However, these finite eluting treatments are generally short-acting and may harm eukaryotic cells, prompting attempts to find safer and longer-lasting, non-eluting solution [32,33]. Following the discovery that mechano-bactericidal nanostructures found on the wing of a dragonfly can kill a variety of bacterial species upon contact [34], it has been shown that titanium surfaces with a specific nanoarchitecture mimicking that of the dragonfly wing can also be highly effective in killing bacteria [34–36]. Mechano-bactericidal activity of nanostructured surfaces is governed by surface profile and is independent of surface chemistry [37]. Nanostructures with a high aspect ratio - with a length many times that of their width, mechanically disrupt the bacterial membrane [38]. Whilst nanostructured surfaces have shown antibacterial efficacy against Gram-positive species, Gram-negative species such as *P. aeruginosa* and *E. coli* are particularly vulnerable, due to differences in cell wall architecture [35,39]. Several methods have been developed to fabricate nanostructures on titanium and its alloys, with antibactericidal properties to enhance osteointegration. Some of these methods include; anodization [40], sol-gel [41] and hydrothermal etched titanium (HTE) [42]. HTE stands out as the method of choice to fabricate $TiO_2$ due to; its simplicity, inexpensive, environmentally friendly, and more importantly can generate a variety of nanoscale patterns [42]. Furthermore, $TiO_2$ surface layer exhibits low solubility in water, which prevents substrate metal ions from dissolution, enhanced wear and fatigue resistance [43].

A limitation of previous studies has been the short time frame over which bacteria were cultured on nanostructured surfaces, whereas biofilms are known to evolve over days and indeed weeks [36,39,44–46]. Moreover, reduced early bacterial viability as well as inhibited and/or delayed biofilm formation on the implant surface may promote improved host tissue integration in the early phase, which further reduces the likelihood of an IAI developing [47,48]. The aim of this study was, therefore, to examine the antibacterial efficacy of a nanostructured titanium alloy on two clinically relevant pathogens (*S. aureus* and *P. aeruginosa*) over a 21-day period. We hypothesise that, there will be a continuous, long-term disruption of biofilm growth and maturation on the antibacterial nanostructured implant surface, which may translate to implants with reduced occurrence of IAIs.

## 2. Materials and methods

### 2.1. Fabrication of antibacterial Ti6Al4V surface

To create the study HTE surface, medical grade Ti6Al4V discs (10 mm diameter, 3 mm thickness) with a mirror polished top surface and a RA of 0.1 μm (Hamagawa Industrial (M)76 SDN BHD, Kedah Malaysia), were hydrothermally etched, using 1 M KOH in a stainless-steel reactor (autoclave; Parr Instrument Company, USA). After the chemical reaction, the vessels were cooled, and samples were cleaned in ultrapure water. After drying for 2 h at 70 °C, discs were heat treated inside a tubular furnace and cooled down overnight. As-received Ti6Al4V discs with a polished top surface were used as the control implant surface (CTL) in biofilm experiments. All samples were cleaned and sterilised at 121 °C for 20 min prior to use in biological experiments.

### 2.2. Characterisation of nanotopography by scanning electron microscopy (SEM) and atomic force microscopy (AFM)

The morphology of the HTE and CTL surfaces was characterised using scanning electron microscopy (SEM) (Zeiss Merlin FEG-SEM, Zeiss, Jena, Germany) at 2 KV, 4.5 mm working distances with magnifications from 5 K to 50 K. Substrate surfaces were orientated at 45° relative to the horizontal plane in the SEM, where an oblique view reveals additional information about the nanotopography that cannot be assessed from a top-down perspective, such as approximate nanostructure height. Five samples were used to measure height, diameter and spacing between nanostructures. The height of the nanostructures was calculated using the distance between a perpendicular plane and the highest point of each spike, and the diameter was measured at mid-height in parallel orientation with the basal plane (25 spikes measured per sample). The measurements for spike height were corrected for the 45° stage tilt [49]. To determine the spacing between nanostructures, zero-degree tilted SEM images were analysed (n = 5), considering nanostructure tips in a 5 μm² field. Analysis was performed using ImageJ software v1.53a (NIH, USA). The morphology of the HTE and CTL surfaces was also characterised in air, using atomic force microscopy (AFM). Data for 5 μm² surface area scans was acquired using a JPK NanoWizard III (JPK BioAFM, Berlin, Germany) with instrument-specific software v5, written in Linux Ubuntu (Canonical Ltd., London, UK). An NT-MDT NSG03 silicon nitride cantilever with a conical tip rated by the manufacturer (Spectrum Instruments Ltd, Moscow, Russia) at a radius less than 10 nm and a half side angle of 18° was used to perform amplitude modulation or tapping mode scans on HTE and CTL surfaces (n = 3). Initial calibration of the cantilever on a glass microscope slide derived a normal spring constant of 1.9 N/m and a deflection sensitivity of 86.8 kHz. Scanning parameters over a scan rate of 0.8 Hz were held constant at a setpoint of 22.8 nm and a drive amplitude of 0.3 V. Roughness values and 3D reconstructions were acquired using Gwyddion data analysis software version 2.54. (http://gwyddion.net/).

### 2.3. Contact angle analysis

The water contact angle (WCA) of the CTL and HTE samples (n = 3) was measured using a contact angle goniometer model RD-SDM02 (RD Support, Scotland, UK), to determine surface wettability. Contact angles were measured for five randomly placed drops on triplicate samples. Both CTL and HTE samples were placed on a flat surface to receive a 4 μL of Milli-Q water and the WCA was measured within 20 s using the sessile drop analysis.

### 2.4. Chemical analysis of the surface by X-ray photoelectron spectroscopy (XPS)

XPS was employed to investigate the surface chemistry of the CTL and HTE samples (n = 3). Survey spectra were collected with a pass energy of 160 eV and high-resolution spectra were obtained using a 20-eV pass energy, with a monochromatic Al source run at 15 keV and 15 mA on a Kratos AXIS Ultra DLD spectrometer (Kratos Analytical Ltd, Manchester, UK). Survey spectra were documented over a 0–1100 eV range at 0.5 eV increments. CasaXPS software version 2.3.23 (www.casaxps.com, Teignmouth, UK) was used to process the spectra after curve fitting, and all binding energies were referenced to the carbon peak at 285.0 eV.

### 2.5. Energy-dispersive X-ray spectroscopy (EDS) for elemental analysis

HTE samples were dry cut with a CBN blade from More Superhard (Zhengzhou, China) at 300 rpm in a Struers Minitome (Willich, Germany). To compare the qualitative elemental analysis between the HTE surface and the subsurface, a Zeiss Merlin FEG-SEM (Zeiss, Jena, Germany) equipped with an EDS detector (Oxford Instruments X-MaxN (20 mm²) EDS-Silicon Drift Detector) was used. Spectra were obtained, using

a SE2 detector at 15 kV, 3 nA, working distance 10 mm, analytic mode and 10 K magnification, using AZTEC EDS software version 3.1 (Oxford Instruments, Oxfordshire, UK).

### 2.6. Short-term cytocompatibility of the HTE nanostructured surface

The ability of RAW 264.7 macrophage like cells (ATCC® TIB-71™, VA, USA) to adhere and proliferate on the HTE surface was assessed by LIVE/DEAD® Viability/Cytotoxicity Kit for mammalian cells (Molecular Probes, Invitrogen, CA, USA), following the manufacturer's protocol. RAW 264.7 macrophage like cells were cultured in Dulbecco's modified Eagle's medium (DMEM, ThermoFisher, MA, USA) supplemented with 10% fetal bovine serum (FCS; ThermoFisher, MA, USA) and 1% penicillin/streptomycin (ThermoFisher, MA, USA). Cells were seeded on HTE and CTL samples at $2.5 \times 10^4$ cells per well in a 48-well tissue culture plate and incubated at 37 °C in 95% humidity and 5% $CO_2$. After 48 h culturing on HTE and CTL samples in a 48-well tissue culture plate, the cells were washed gently with phosphate buffered saline (PBS), followed by calcein AM/EthD-1 working solution and incubated at room temperature for 30 min, then imaged by confocal laser scanning microscope (CLSM; Olympus, Tokyo, Japan). Calcein (Ex/Em 495/515 nm) will fluorescently stain live cells green, whereas EthD-1 (Ex/EM 495/635 nm) will stain cells with damaged membranes or dead cells, producing a bright red fluorescence upon binding to nucleic acids. Viability was quantified by counting red and green stained cells using ImageJ software v1.53a (NIH, USA). Cytotoxicity was further assessed using LDH-Glo™ Cytotoxicity Assay (Promega, WI, USA), to detect extracellular lactate dehydrogenase (LDH), a widely used marker in cytotoxicity studies due to its rapid release in tissue culture media upon disruption of the cell membrane [50,51]. Briefly to quantify LDH release, 50 μL of media was added to 50 μL of LDH detection reagent in a 96-well opaque plate, mixed and incubated at room temperature for 30 min then luminescence read on a Synergy HTX multi-mode microplate reader (Biotek, VT, USA). Triplicate readings were normalised to cells grown on tissue cultures plate (TCP). The formula used to calculate percent cytotoxicity was,

$$\% \, Cytotoxicity = \frac{Treatment \, LDH \, Release - Background}{Control \, LDH \, Release - Background} x \, 100$$

### 2.7. Bacterial culturing and biofilm growth protocols

*P. aeruginosa*, ATCC 15692 and *S. aureus*, ATCC 25923 glycerol stocks were plated onto Tryptone Soy Agar (TSA; Oxoid, ThermoFisher, MA, USA) plates and incubated for 18 h at 37 °C. Single colonies of each species were inoculated into Tryptone Soy Broth (TSB; Oxoid, Thermo-Fisher, MA, USA) and 5% FCS (TSBFCS; Gibco, ThermoFisher, MA, USA), and incubated overnight at 37 °C. The following day bacterial cultures were gram stained to confirm purity and adjusted to $1 \times 10^6$ CFU/mL by adjusting to the associated 600 nm ($OD_{600}$) reading measured on a Nanodrop 2000C (ThermoFisher Scientific, MA, USA). An $OD_{600}$ of 1 was determined to be approximately $1 \times 10^9$ CFU/mL by calibrating against colony forming units (CFU) for *P. aeruginosa* and *S. aureus* prior to the commencement of this study. Samples of HTE and CTL were inoculated in a 24-well plate with 1 mL of $1 \times 10^6$ CFU/mL of either *P. aeruginosa* or *S. aureus* for nine time points (day 1–7, 14 and 21) and incubated with daily replenishment of TSBFCS, at 37 °C in a humid chamber on an orbital shaker (Ratek Instruments, VIC. Australia), on setting 6. At each time point HTE and CTL samples were removed for analysis.

### 2.8. Quantitative antimicrobial test

The gold standard for enumerating bacteria is the colony forming unit (CFU) method [52]. Briefly, at each time-point (day 1–7, 14 and 21), HTE and CTL discs were transferred into 1 mL PBS and vortexed for 15 s, followed by 2 min sonication then vortexed a further 15 s, and serially diluted. Serial dilutions were plated onto TSA plates and incubated for

18 h at 37 °C. Viable cell counts were performed on the HTE and CTL surfaces by the drop plate method [50]. The following day, colonies were counted, and CFUs determined for *P. aeruginosa* and *S. aureus* incubated on HTE and CTL discs [53]. Comparison between viability on the HTE and CTL samples was further used to determine log reduction, $\log_{10}$ ( $N_{CTL} / N_{HTE}$ ) and percent reduction for each time-point. To validate complete removal of bacteria the HTE and CTL samples were imaged by SEM, post CFU processing.

### 2.9. Bacteria live/dead assay

Bacterial cells attached to the surface of HTE, and CTL samples were stained by Live/Dead® BacLight™ (Invitrogen, ThermoFisher MA, USA) according to the manufacturer's protocol. Briefly, a mixture containing Syto9 (Ex/Em 480/500 nm) and Propidium Iodide (PI; Ex/Em 490/635 nm) in equal proportions (1.5 μL of each per mL) in PBS was freshly prepared. Syto9 is taken up by all cells and binds with nucleic acids, staining the cell green. Whereas PI is unable to pass through intact plasma membrane and is therefore only taken up by damaged cells allowing detection of both live (green) and compromised cell membrane integrity that is associated with cell death (red). The samples were immersed in the live/dead solution and incubated at room temperature for 15 min in the dark prior to microscopy.

### 2.10. Confocal laser scanning microscope analysis

Samples were imaged at each time-point (days 1–7, 14 and 21), using three random images per sample. CTL and HTE samples were imaged for viability and biomass analysis, using an Olympus FV3000 CLSM (Olympus, Tokyo, Japan). Single plane micrographs were used to determine viability by counting red and green stained cells using ImageJ software v1.53a (NIH, USA). Z-stack images were acquired for *S. aureus* and *P. aeruginosa* on days 7, 14 and 21. The viability and total biomass for the Z-stacks were also examined using Imaris 3D analysis software (Version 9.3.0, Bitplane, Zürich, Switzerland). The excitation/emission maxima of the LIVE/DEAD Bacterial Viability Kit dyes are 480/500 nm for SYTO 9 and 490/635 nm for PI.

### 2.11. SEM analysis of bacterial morphology

To prepare biological samples for SEM, samples were fixed in 1.25% glutaraldehyde, 4% paraformaldehyde in PBS containing 4% sucrose for 24 h. Next, they were washed in PBS, followed by dehydration in an ascending ethanol series from 50%, 70% and 100% ethanol. Finally, samples were chemically dried using hexamethyldisilane (Sigma-Aldrich, MO, USA), mounted on aluminium stubs, sputter coated with 1 nm platinum, and imaged using a Zeiss Merlin FEG- SEM (Carl Zeiss, Jena, Germany).

### 2.12. Statistical analysis

Graphical data was represented by mean and standard deviation. All statistical analysis were performed using GraphPad Prism version 9.0.0.0 for Windows (GraphPad Software, La Jolla California USA, www.graphpad.com). The data was analysed by multiple t-tests. Statistical significance was determined using the Holm-Sidak method, with alpha = 0.05, significance levels were set at p < 0.05. All experimental procedures were performed in biological replicates (n = 3).

## 3. Results and discussion

### 3.1. Characterisation of HTE nanostructure surface

Representative SEM micrographs depicting the nanostructured features of the CTL and HTE samples are shown in Fig. 1A–D at 5, 50 and 200 K magnification, respectively. The CTL surface appeared smooth at
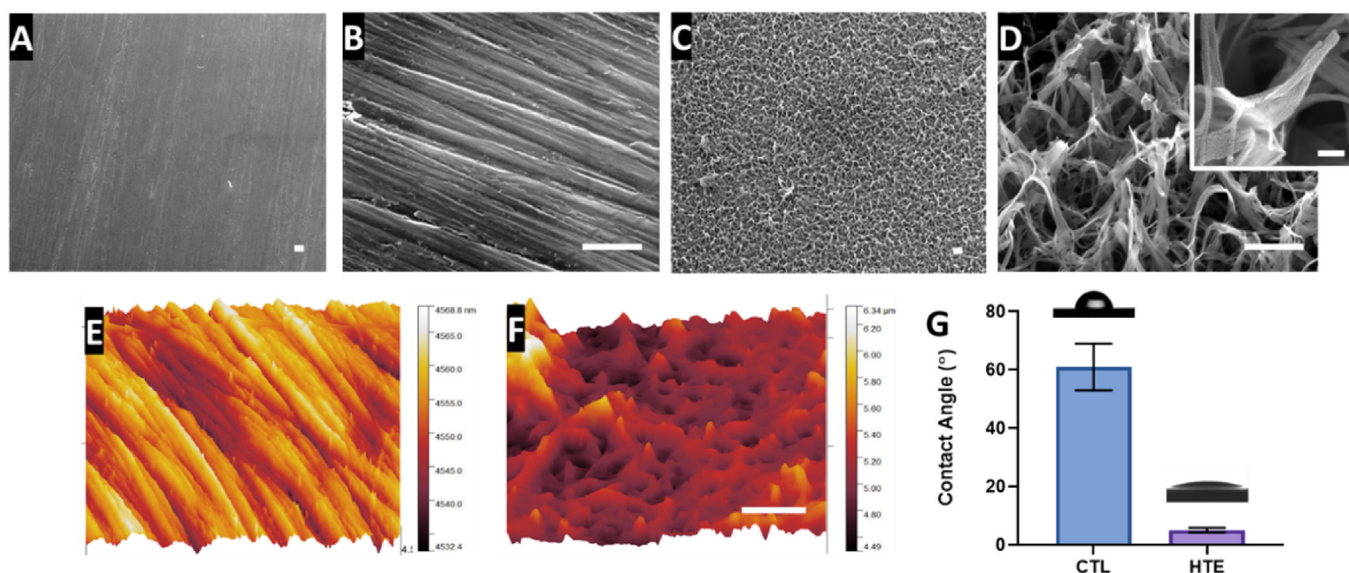
**Fig. 1.** SEM micrographs of unetched CTL (1A and 1B) showing effects of polishing; and the HTE surface (1C and 1 D) displaying hierarchal nanotopography (scale bar for Fig. 1A–D = 500 nm). The inset image in 1 D shows the morphology of a single nanostructure at 200 K magnification (scale bar = 100 nm). AFM-3D reconstructions showing nanotopological features of a 5 × 5 μm region on CTL (1E) and the superhydrophilic HTE surface (1F) (scale bar = 1 μm). The bar graph represents the water contact angle for CTL and HTE surfaces mean ± SD (n = 3), p = 0.0003. Images above the bar graph demonstrate the wettability by the sessile drop method (1G). Nanostructure measurements, AFM and wettability can be found in supplementary information Table S1.

lower magnification, however, at high magnification, machining marks and scratches due to the polishing process can be seen. The HTE surface exhibited hierarchal disordered spike-like nanostructures covering the entire surface. The height of the nanostructures was $348 \pm 152$ nm, and diameter at mid-height was $98 \pm 60$ nm, signifying a high aspect ratio. The spacing between the individual nanostructures was $437 \pm 46$ nm. Nanostructures with similar dimensions have previously been shown to be effective at killing both Gram-positive and Gram-negative bacteria [35,38].

The AFM image, presented in Fig. 1E, confirmed machining marks due to polishing on the CTL surface. Nanostructures with peaks and valleys can also be seen in the AFM image from the HTE surface (Fig. 1F), however the fine details seen by SEM could not be resolved by AFM due to the well-known tip convolution effect [54]. The AFM images displayed greater roughness on the HTE surface compared to the CTL surface. The root mean square roughness (RMS), arithmetic roughness average (Ra) and surface area (SA) determined from the AFM images for the CTL surface were 5.13 nm, 4.15 nm and 25.14 $\mu m^2$, respectively, while for the HTE surface these parameters were 236.3 nm, 175.5 nm and 64.64 $\mu m^2$ respectively. The surface area measured by AFM on the HTE sample increased by 2.6-fold when compared to the CTL. It should be noted, however, that the tip convolution resulting from the cantilever tip side angle and scan velocity reduced the measured surface roughness values, especially towards surface features with comparable magnitude to the tip radius. The water contact angle for the control surface ($61 \pm 8°$) was significantly greater than that of the HTE surface ($5 \pm 0.8°$, p = 0.003, Fig. 1G). The Wenzel [55] and Cassie–Baxter [56] equations state that adding surface roughness will enhance the wettability caused by the chemistry of the surface. It is well known a water contact angle between $0°$ and $10°$ is categorised as superhydrophilic [57]. Generally, an increase in hydrophilicity is an imperative characteristic to improve biocompatibility and antibactericidal properties of the HTE surface - where hydrophilic implant surfaces favour cell-implant contact, the initial phases of wound healing and a cascade of events leading to the osteogenesis [58–61].

### 3.2. Chemical analysis

The atomic percentage of elements detected on the CTL and HTE disc surfaces using XPS is shown in Fig. 2A. There was a significant increase in oxygen and titanium atomic concentrations on the HTE surfaces compared to the untreated CTL (5.6%, p = 0.0003 and 5.1%, p = 0.0002 respectively). These results are consistent with formation of a thicker oxide layer occurring during the hydrothermal etching process. The increased thickness of the oxide layer also meant that, with the XPS sampling depth of 10 nm, vanadium and aluminium became barely detectable on HTE surfaces. Hydrothermal etching also reduced the presence of environmental contaminants such as carbon and nitrogen. Furthermore, a small amount of potassium was incorporated within the HTE surface, consistent with the use of KOH as the alkaline etchant. Trace amounts of magnesium, calcium, and silicon, incorporated during polishing and handling of the CTL samples, were detected on both CTL and HTE disc surfaces.

Using EDS (Fig. 2B and C), we confirmed the increase in oxygen content for the HTE surface, consistent with hydrothermal oxidation of the surface layer occurring during hydrothermal etching discussed above [42]. The three elements used in the manufacture of the Ti-alloy (i.e., Ti, Al, V) were also detectable in the EDS spectra.

### 3.3. In vitro cytotoxicity of raw 264.7 cells to HTE treated surface

After biomaterial implantation, macrophages are one of the earliest host-cell responders [62]. We therefore used a RAW 264.7 macrophage-like, Abelson leukemia virus-transformed cell line derived from mice, to observe any short-term cytocompatibility to the HTE surface. Briefly, the viability of RAW 264.7 cells seeded onto CTL and HTE discs are shown in Fig. 3A. After 48 h incubation, the percent viability for CTL and HTE discs was $95.8 \pm 2.6\%$ and $94.0 \pm 4.2\%$, respectively (p = 0.57). Furthermore, after fluorescent staining with LIVE/DEAD® Viability/Cytotoxicity Kit, no red stained cells (indicating dead cells) were detected on the CTL or the HTE samples. Cells on both CTL and HTE samples appeared healthy and had comparable morphology. The cells were approximately 60% confluent after 48 h on both CTL (Fig. 3B) and HTE samples (Fig. 3C).
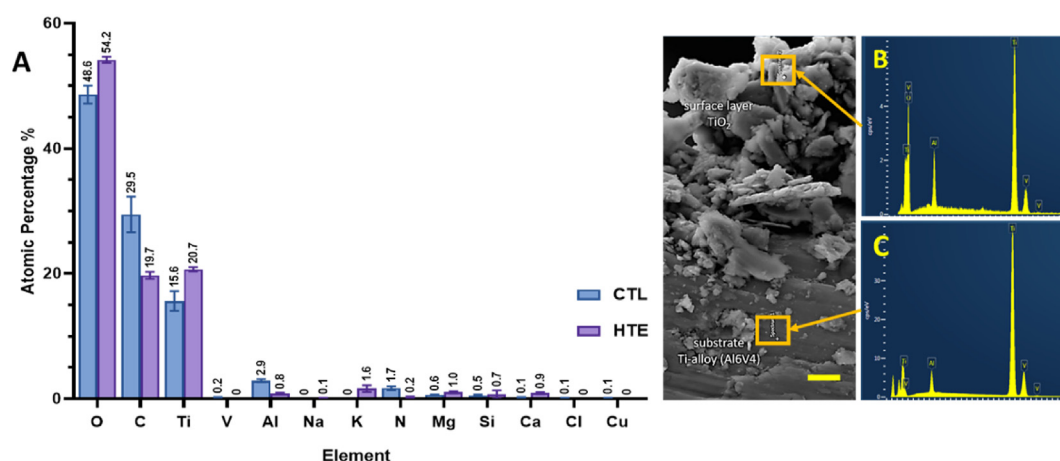
**Fig. 2.** Atomic percentages for CTL and HTE samples quantified by XPS, mean ± SD (n = 3). p-values for all elements detected can be found in supplementary information in Table S2 (2A). EDS spectra measured on HTE surface (2B) and substrate (2C) (scale bar = 5 μm).
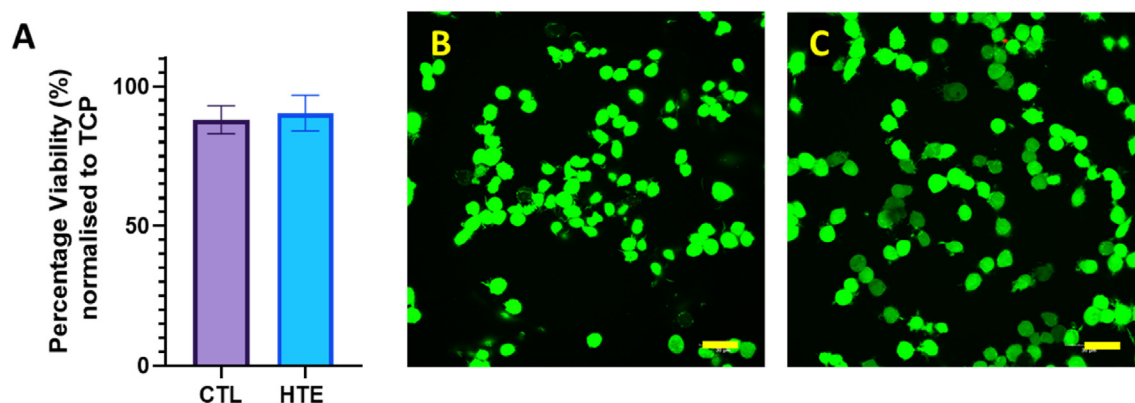


**Fig. 3.** Viability of RAW 264.7 macrophage-like cells determined by LDH-Glo™ Cytotoxicity Assay, normalised to RAW 264.7 macrophages grown directly on TCP. The mean ± SD percentage viability for RAW 264.7 macrophages incubated on CTL samples was 95.8 ± 2.6% and HTE samples was 94.0 ± 4.2% (mean ± SD), p = 0.57, n = 3 (3A). The confocal images of RAW 264.7 macrophages incubated on the CTL (3B) and HTE (3C) for 48 h, stained with LIVE/DEAD™ Viability/Cytotoxicity Kit, for mammalian cells (scale bars = 30 μm).
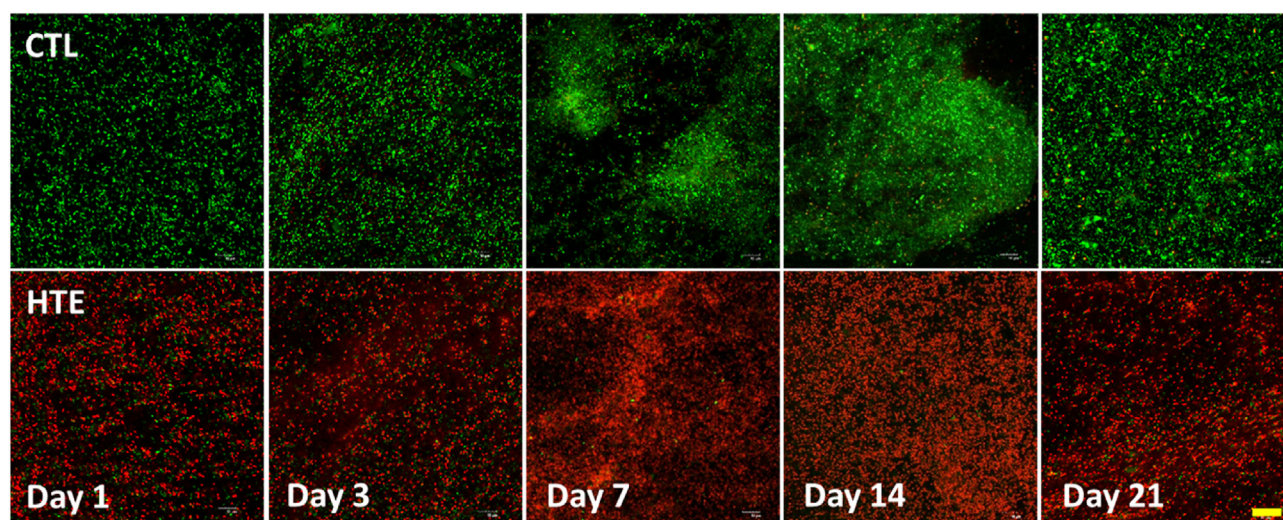


**Fig. 4.** Confocal microscopy images of *P. aeruginosa* after staining with LIVE/DEAD® BacLight™ Bacterial Viability Kit incubated on CTL and HTE surfaces at five representative time points over 21-days. Live cells stain fluorescent green, whereas bacteria with damaged membranes stain fluorescent red. Top panels represent as-received CTL, and the bottom panels represent the matching time point HTE samples incubated with *P. aeruginosa* (n = 3). Scale bar = 20 μm. (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)

## 3.4. Bacterial viability analysis

To assess the viability of bacterial populations on the CTL and HTE surfaces, we used LIVE/DEAD® BacLight™ Bacterial Viability Kit [63]. Representative live/dead images of *P. aeruginosa* and *S. aureus* at days 1, 3, 7, 14 and 21 are shown in Figs. 4 and 5, respectively. When *P. aeruginosa* was cultured on HTE surfaces, the bacterial viability was only $0.97 \pm 0.06\%$ after 24 h. Longer incubation times resulted in a marginally higher mean viability, which never exceeded 4%. (Fig. 6A). For *S. aureus*, there was an initial cell viability of $26.0 \pm 0.9\%$ after 24 h incubation. On day 3, *S. aureus* viability was at its lowest ($6.8 \pm 3.1\%$), with viability increasing to $36.2 \pm 2.7\%$ by day 7, and then $40.9 \pm 4.3\%$, by day 21 (Fig. 6B). In contrast, the viability for *S. aureus* and *P. aeruginosa* did not drop below 90% throughout the 21-day study on the CTL surface (Fig. 6A and B).

To analyse the biovolume on the CTL and HTE surfaces, we prepared Z-stack surface rendered 3-D models for *P. aeruginosa* (Fig. 7) and *S. aureus* at days 7, 14 and 21 (Fig. 8). Bacterial viability was also quantified at the three time points (day 7, 14 and 21) using 3D rendered biovolume Z-stack images obtained from the same samples used for single-plane fluorescence imaging. Each of the three time points represented an additional week of biofilm colonisation for *P. aeruginosa* (Fig. 7H) and *S. aureus* (Fig. 8H). *P. aeruginosa* incubated on the CTL samples showed significantly greater biovolume compared to the HTE samples at days 7, 14 and 21 ($p = 0.003$, $p = 0.006$ and $p = 0.003$ respectively, Fig. 7G). There was minimal evidence of clusters of live bacteria, associated with the formation of biofilm [64], on the HTE surfaces. Rather, there were thick areas of dead bacteria evident on these samples, most prominent at days 14 and 21. Although there was some increase in *P. aeruginosa* biovolume on the HTE surface by day 21, this was still less than half the biovolume on the CTL surface at day 7. While the viability of *P. aeruginosa* remained above 90% on the CTL surfaces out to day 21, from day 7 to day 14, the viability on the HTE surface remained below 25% ($21.1 \pm 3.7\%$ at day 7, decreasing to $18.1 \pm 3.5\%$ at day 14), with most cells in contact with the surface killed at day 21. Interestingly, there was evidence of minor bacterial recolonisation above the previously killed bacteria on the HTE surface, which may partially explain the slight recovery in *P. aeruginosa* viability through the biovolume to $42.4 \pm 5.2\%$ by day 21 (Fig. 7H). Similarly, to *P. aeruginosa*, there was significantly less *S. aureus* biovolume on the HTE surfaces compared to the CTL surfaces at days 7, 14 and 21 ($p = 0.002$, $p = 0.01$

and $p = 0.01$ respectively, Fig. 8G). Again, by day 21, there was less than half the biovolume of *S. aureus* on the HTE surface compared to that on the CTL surface at day 7. Pockets of *S. aureus* recolonisation were evident away from the surface, particularly noticeable on days 14 and 21 (Fig. 8E and F), where total viability increased from $35.0 \pm 6.2\%$ at day 7, $53.1 \pm 3.0\%$ at day 14, to $63.4 \pm 4.2\%$ at day 21 (Fig. 8H).

Using the drop plate method, the HTE surface showed a 1.9–2.5 log reduction in CFUs for *P. aeruginosa*, with the highest log reduction recorded on day 2 (2.5 log reduction). In contrast, *S. aureus* displayed a 1.5–2.5 log reduction in CFUs (Fig. 9A). The maximum log reduction for *S. aureus* on the HTE surface was on day 3, consistent with data from live/dead staining (Fig. 6B).

After day 3, the HTE samples incubated with *S. aureus* showed a consistently lower log reduction compared to *P. aeruginosa* until the end of the study, particularly on days 5, 6 and 14 ($p < 0.05$). Both *P. aeruginosa* and *S. aureus* showed CFU counts in the range of 2–4 orders of magnitude higher on the CTL than on the HTE surface throughout the 21-day study (Fig. 9B and C). There was a notable increase in log reduction on *S. aureus* incubated on HTE samples at day 3, in agreement with live/dead viability (Figs. 5 and 6B). The unexpected finding that *S. aureus* showed a delayed maximum killing on the HTE surface at day 3 could be a function of the difference in membrane structure between Gram-positive and Gram-negative bacteria [65].

## 3.5. Morphology of P. aeruginosa and S. aureus on CTL and HTE surfaces

To gain further insights into differences between the interaction of *P. aeruginosa* and *S. aureus* with CTL and HTE surfaces over time, SEM micrographs were acquired over 21 days. For bacteria incubated on the HTE surface, there were distinct differences in morphology from days 7–21 compared to CTL surfaces. *P. aeruginosa* and *S. aureus* had almost completely colonised the CTL surfaces by day 7, with increasing cell density and little evidence of cell damage by day 21. The morphology of *P. aeruginosa* remained non-deformed, regular rod-shaped, attached to the surface by pili on CTL surfaces (Fig. 10, A-C). Similarly, *S. aureus* incubated on CTL surfaces, appeared spherical, with little evidence of membrane disruption, increasing in density and surface coverage, with notable mushroom-shaped formations indicative of biofilm formation by day 21 (Fig. 11, A-C) [64]. Further evidence that a mature biofilm had formed on the CTL surfaces by day 7 is the extensive extra polymeric substance (EPS) [66], which increased in density by day 21. In contrast,
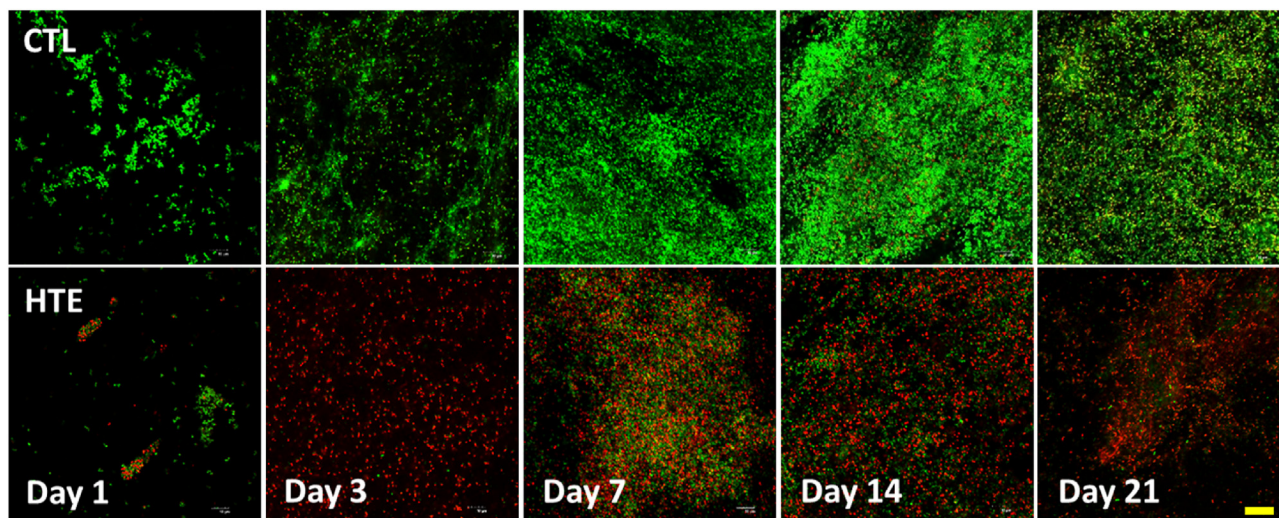


**Fig. 5.** Confocal microscopy images of *S. aureus* after staining with LIVE/DEAD® BacLight™ Bacterial Viability Kit incubated on CTL and HTE at five representative time points over 21 days. Live cells stain fluorescent green, whereas bacteria with damaged membranes stain fluorescent red. The top panels represent as-received CTL, and the bottom panels represent the matching time point HTE samples incubated with *S. aureus* ($n = 3$). Scale bar = 20 μm. (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)
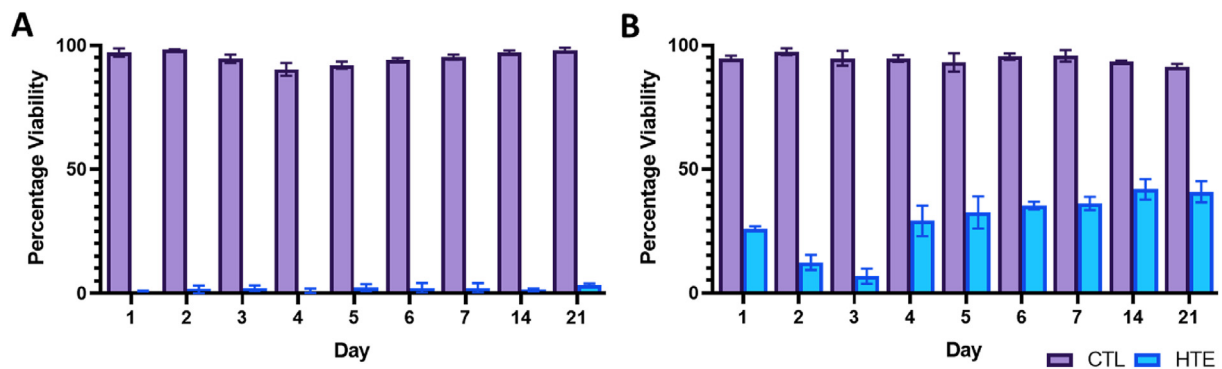
**Fig. 6.** Viability of *P. aeruginosa* (6A) and *S. aureus* (6B) determined from the live/dead staining over 21 days on CTL and HTE surfaces, mean ± SD (n = 3). To compare CTL and HTE surfaces, multiple comparisons using the Holm-Sidak method revealed a significance of p < 0.0001 at all-time points. Table of p-values can be found in the supplementary information (Table S3 and S4 respectively).
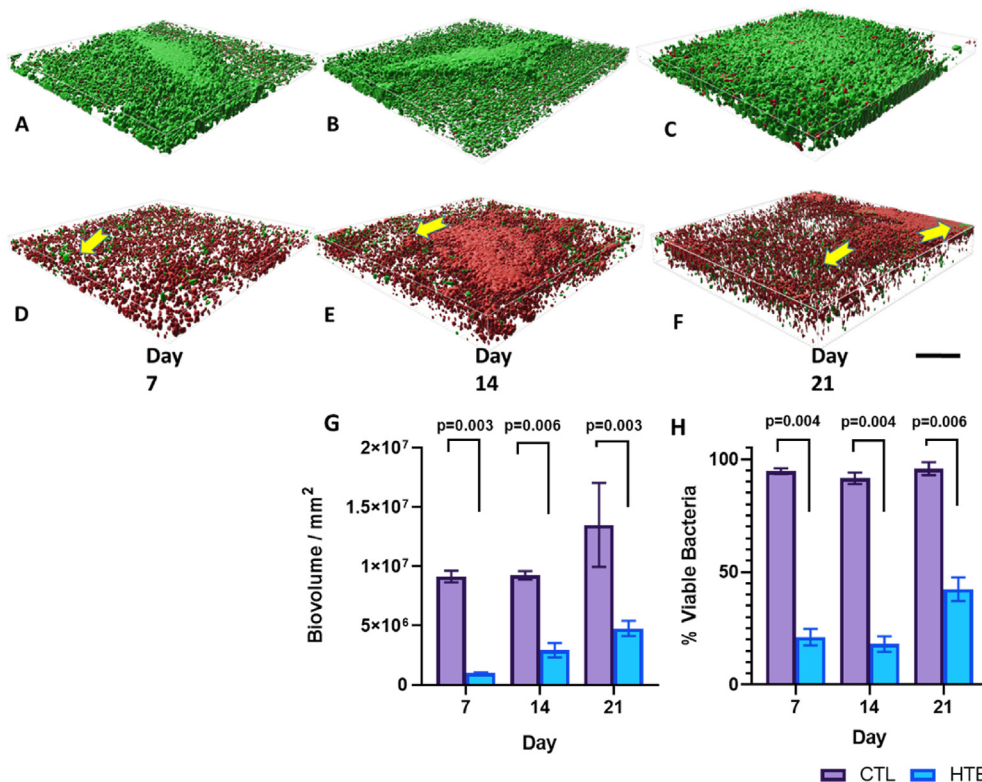


**Fig. 7.** Representative Z-stack rendered biofilm 3D-images of *P. aeruginosa* on CTL and HTE surfaces for day 7 (Fig. 7A and D, respectively), 14 (Fig. 7B and E, respectively) and 21 (Fig. 7C and F, respectively). Biovolume of *P. aeruginosa* incubated on CTL and HTE for day 7, 14 and 21 (Fig. 7G) mean ± SD, (n = 3). Yellow arrows highlight pockets of bacterial recolonisation on top of dead biofilm (scale bar = 30 μm), images signify an area of 0.01 mm² (n = 3). Viability of *P. aeruginosa* (Fig. 7H) at three time points calculated from biovolume using multiple t-tests followed by the Holm-Sidak method, with alpha = 0.05. Each column represents mean ± SD, (n = 3). (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)

both bacterial species remained in small clusters on the HTE surfaces out to day 21, with continued evidence of cell rupture. Examples of lysed, flattened, and fragmented *P. aeruginosa* can be seen in Fig. 10, D-E, and the ruffled and fragmented morphology of *S. aureus* can be seen in Fig. 11, D-E. The inset images in Figure S1 demonstrate membrane piercing by the nanostructures, consistent with other reports where both Gram-negative and Gram-positive bacteria have been killed on comparable surfaces at earlier timepoints [37,67]. As confirmed by live/dead viability assays (Figs. 4 and 6A) and CFU counts (Fig. 9), *P. aeruginosa* appeared to be inflicted with more cellular damage than *S. aureus* due to the nanostructures on the HTE surface. Significantly, we observed minimal EPS produced by both bacteria and fewer cells on the HTE surfaces compared to CTL samples at all timepoints (Figs. 10 and 11, A-C). During biofilm formation EPS is vital to the shift in cell state from adhered planktonic cells to a recalcitrant mature biofilm, where bacteria within the biofilm are more resistant to antibacterial drugs and host immune responses [68,69]. Cao et al. [70] reported similar findings on

nanostructured titanium alloy towards *S. epidermidis*, where biofilm formation was delayed for up to 6 days. However, to the authors' knowledge this is the first study that demonstrates delayed biofilm formation for *P. aeruginosa* and *S. aureus* on a nanostructured surface for 21 days.

Bioinspired nanostructured surfaces have previously been reported to show antibacterial efficacy against pathogens involved in IAIs [34,35,38, 71]. However, this is the first study to examine the efficacy of such a surface for extended periods of time. We evaluated the efficacy of the nanostructures against two pathogens commonly associated with IAIs, i.e., the Gram-negative *P. aeruginosa* and the Gram-positive *S. aureus,* over the period of 21 days. We observed a sustained high killing rate on the HTE surface for both *P. aeruginosa* and *S. aureus* of approximately 97% and 60% by live/dead analysis at day 21, respectively. Analysis of biovolume demonstrated that *P. aeruginosa* and *S. aureus* remained only approximately 60% and 40% viable by day 21, respectively, where most live bacteria were away from the surface. Conversely, *P. aeruginosa* and
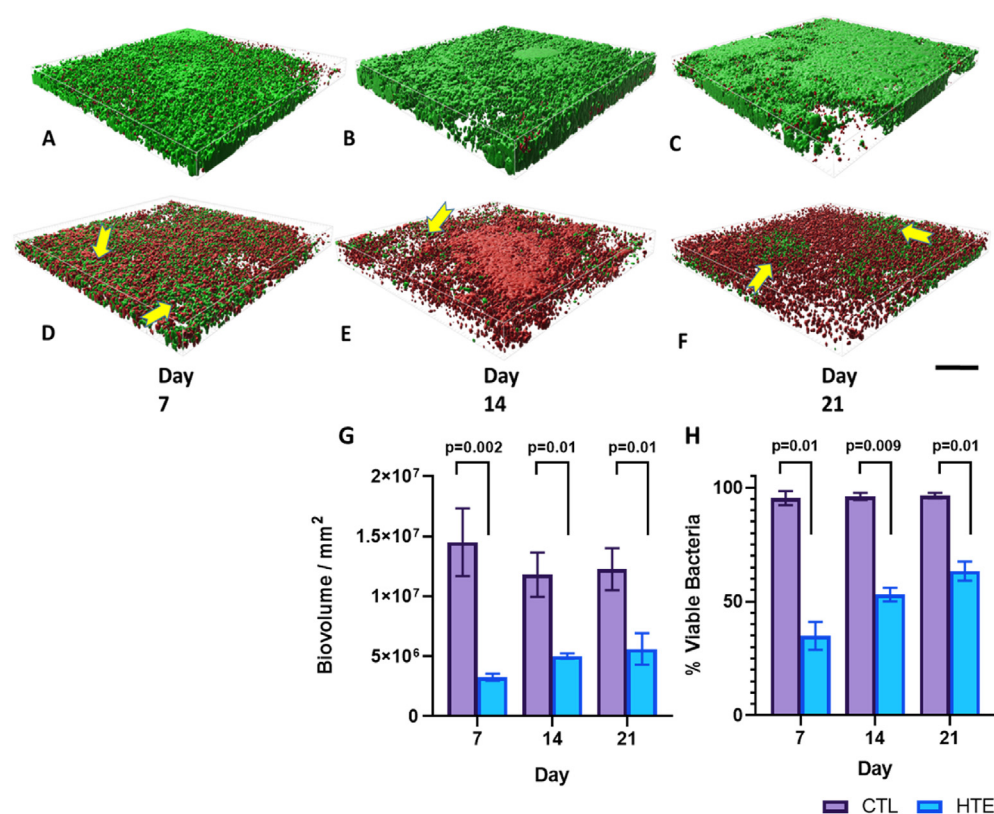
**Fig. 8.** Representative Z-stack rendered biofilm 3D-images of *S. aureus* on CTL and HTE samples for day 7 (Fig. 8A and D, respectively), 14 (Fig. 8B and E, respectively) and 21 (8C and 8F, respectively). Scale bar = 30 μm. Biovolume of *S. aureus* incubated on CTL and HTE for day 7, 14 and 21 (6G) Mean ± SD, n = 3. Yellow arrows depict pockets of bacterial recolonisation on top of dead bacteria (scale bar 30 μm), images signify an area of 0.01 mm² (n = 3). Viability of *S. aureus* (8H) at three time points calculated from biovolume using multiple t-tests followed by the Holm-Sidak method, with alpha = 0.05. Each column represents mean ± SD, (n = 3). (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)



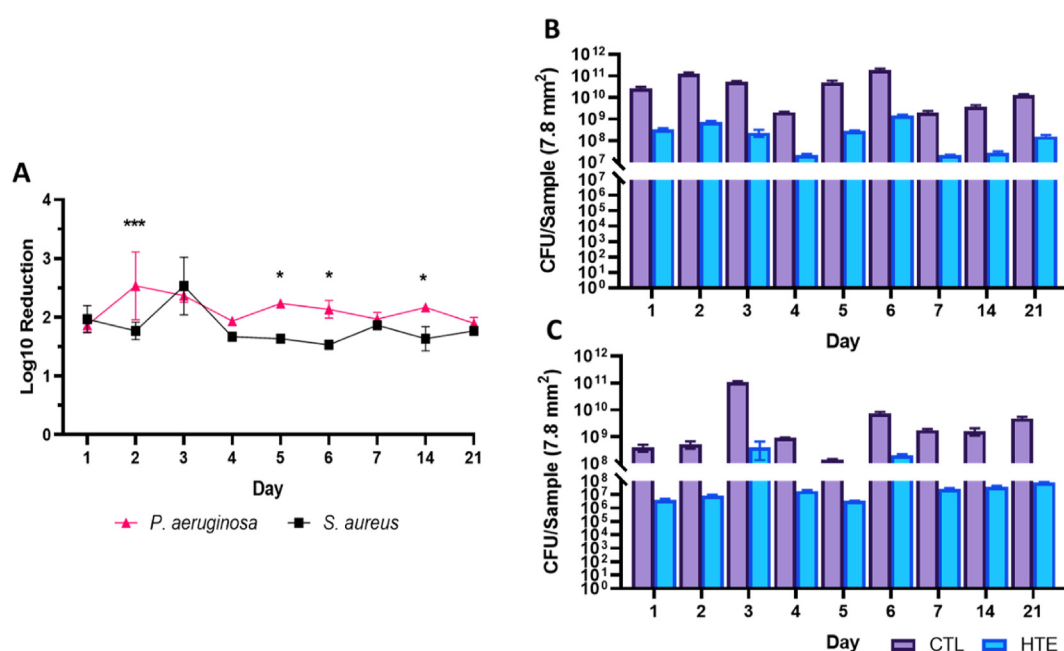**Fig. 9.** Log reduction of *P. aeruginosa* and *S. aureus* incubated on HTE surface over 21 days. Mean ± SD, (n = 3) and * = p < 0.05 and *** = p < 0.001 (9A). CFUs per sample for *P. aeruginosa* (9B) and *S. aureus* (9C) on CTL and HTE surface mean ± SD, (n = 3). Table of p-values can be found for Fig. 9A, B and 9C in the supplementary information (Table S5, Table S6 and Table S7, respectively).

*S. aureus* incubated on CTL surfaces remained greater than 95% viable throughout the study, both at the interface with the surface and throughout the full thickness of the biomass. There was a consistent 2-log reduction in viability for both bacteria on the HTE surface compared to CTL throughout the study. SEM images provided further evidence of a mechano-bactericidal effect of the spike-like nanostructures, also shown by other researchers at early timepoints [37,67], however our study showed that this effect was prolonged, with far fewer bacteria colonising the surface on the HTE samples compared to CTL samples out to day 21. By day 21, there remained a lack of EPS on HTE surfaces, where flattened, disrupted, and fragmented bacteria were observed. In contrast, both bacteria appeared to have formed mature biofilms on the CTL
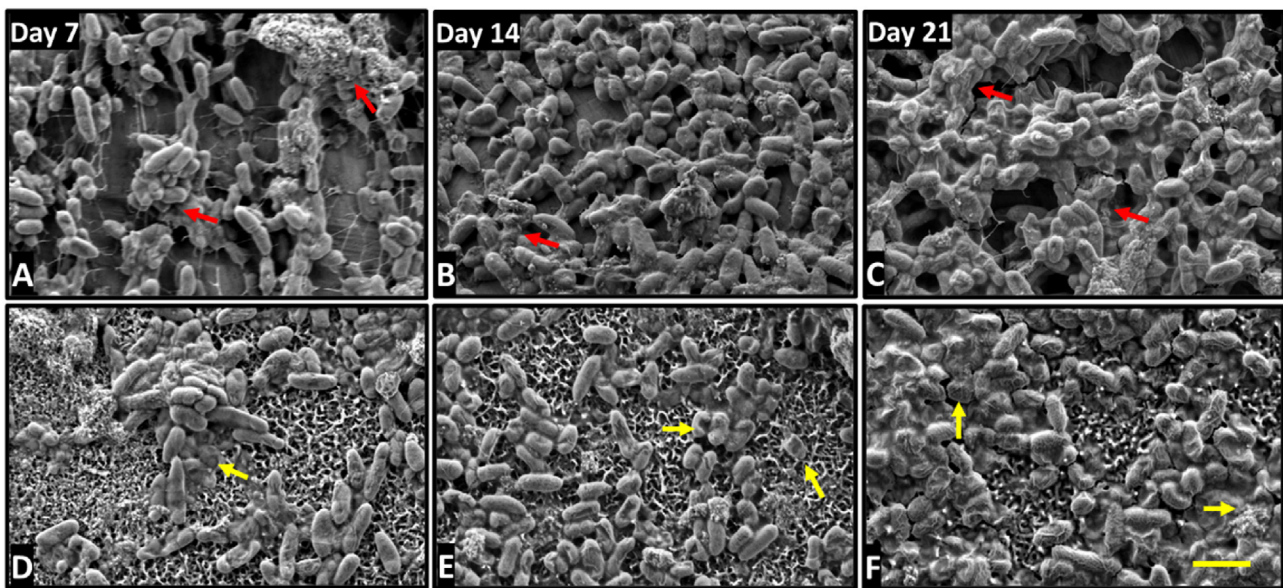
**Fig. 10.** SEM micrographs of *P. aeruginosa* incubated for 7, 14 and 21 days on CTL (A–C) and HTE surfaces (D–F). Yellow arrows are highlighting irregular morphology and dead cells: Red arrows show EPS covering bacteria. Scale bar bottom left = 2 μm. (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)
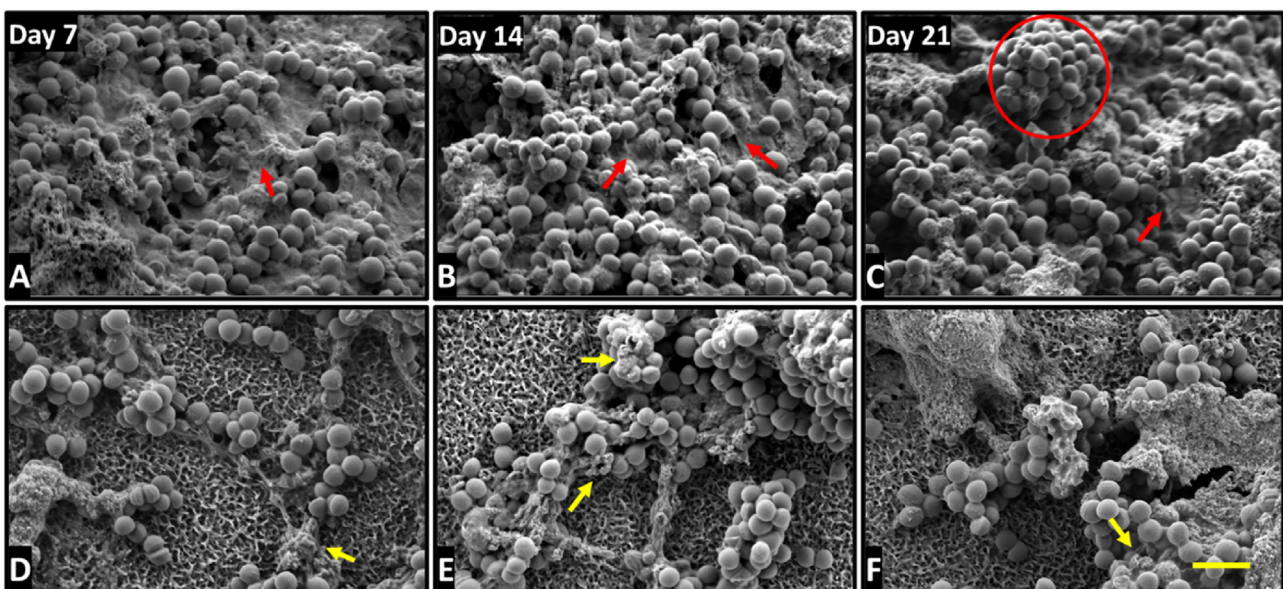


**Fig. 11.** SEM micrographs of *S. aureus* incubated for 7, 14 and 21 days on CTL (A–C) and HTE surfaces (D–F). Arrows are highlighting irregular morphology and dead cells; Red arrows show EPS covering bacteria: Red circle highlights the formation of mushroom-shaped biofilm. Scale bar bottom left = 2 μm. (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)

surfaces. Taken together, these results confirm our hypothesis that the antibacterial nanostructured implant surface disrupted biofilm growth and maturation over a long period.

Although the HTE surface was effective at killing both bacteria in this study, it appeared more so against *P. aeruginosa*, which may reflect the differences between Gram-negative and Gram-positive bacteria. It is well known that Gram-negative bacteria are less rigid than their Gram-positive counterparts, predominately due to the thinner peptidoglycan layer present in the cell wall, thus rendering them vulnerable to contact killing by the spike-like nanostructures [65,72]. Although Gram-positive bacteria lack an outer membrane, they are characterised by a thick peptidoglycan layer and protective capsular polysaccharide outer layer, providing greater cell wall rigidity and increased resistance to

mechanically induced killing than Gram-negative bacteria [73,74].

The rate of cell division may also be a vital contributing factor to the difference in killing rate observed between Gram-negative and Gram-positive bacteria on the HTE surface. During cytokinesis, a complex hierarchical assembly of proteins form at the mid-cell known as the divisome [75]. This remodelling may alter the bacterial turgor pressure [76, 77] and cell wall integrity [78,79], resulting in heightened killing on the HTE surface of bacteria which divide more rapidly. The doubling time for *P. aeruginosa* can be as low as 25 min [80], in contrast to 1 h for *S. aureus* [81]. The differences in doubling time between the two bacteria may be an additional influence on the killing rate on the HTE surface, reported by in this study and by other researchers [36,42,44,82].

Importantly, we observed substantial volumes of dead bacteria

throughout the biomass, away from the HTE surface over 21 days. It is likely that not all these dead bacteria visible throughout the biomass came directly into contact with the nanostructured surface. Stress-induced programmed cell death (PCD) may provide an explanation for how a passive non-eluting surface may induce bacterial cell death at some distance away from the surface. PCD refers to the genetically encoded cascade of events leading to cell death, and may be initiated by cellular stress, for example oxidative stress [83,84]. Although PCD is a well understood phenomenon within eukaryotic cells, there are emerging studies that show PCD has been conserved throughout evolution in prokaryotic cells [85,86]. A biofilm may induce PCD within cells that are undergoing excessive environmental stresses, such as contact killing with a nanostructured surface, to favour the survival of the colony [86,87]. While the majority of the biovolume appeared to be constituted from dead bacteria on the HTE surfaces, there were observable pockets of live cells remerging on top of previously killed cells, particularly evident for *S. aureus* at day 14 and 21. Additionally, stressed cells in contact with the HTE surface may express extracellular death factor (EDF), which has been implicated in PCD in Gram-negative bacteria, mainly studied in *E. coli* [88,89].

Furthermore, gene expression may be modulated by the nano-structures, perhaps resulting in biofilm disaggregation. Quorum sensing genes, which synchronise the biofilm allowing it to behave like a multicellular organism, may be modulated, thus affecting biofilm for-mation over time. Potential down regulation in quorum sensing and biofilm associated genes may inhibit formation of biofilm on the HTE surface, though this warrants further investigation. It is obvious that unravelling a complete mechanistic picture of the effect of surface nanostructures on bacterial attachment, growth, colonisation, and bio-film establishment is a complex task that will require the efforts of much ongoing research. However, this work details pioneering experiments which demonstrate the capacity of surfaces containing disordered spike-like nanostructures to sustain antibacterial properties over an extended period. Clinically, such an implant surface may not only act as a deterrent to bacterial aggregation and biofilm formation, but it may also provide host tissue with more time to win the race to the surface, in turn reducing the likelihood of an IAI.

## 4. Conclusion

In this study, we characterised the physico-chemical and antibacterial properties of hydrothermally etched nanostructures over extended time periods. A short-term cytocompatibility study using RAW 264.7 macrophage-like cells showed no adverse effects on morphology, viability, and proliferation. Following culture on HTE surfaces, *P. aeruginosa* did not exceed 4% viability over a 21-day period, whereas *S. aureus* viability reduced to 6.8 ± 3.1% by day 3, then increased to approximately 40% by day 21. In contrast, both bacterial species remained above 95% viable on CTL surfaces over the duration of the study. Total biovolume and measured viability was also significantly lower on the HTE surfaces for both bacteria at all timepoints. SEM im-aging confirmed contact killing of bacteria on the nanostructures, as well as a lack of mature, organised biofilm forming on the HTE surfaces over longer timepoints. Although additional research is required to fully un-derstand the mechanisms underpinning the long-term antibacterial properties of surfaces containing sharp nanostructures, including an *in vivo* model, this study points to the potential of such surfaces to be applied to medical devices and ultimately contribute to reducing the rates of IAI.

## Author contributions

R.B., and D.F. designed, planned, and performed the experiments. R.B. analysed data and drafted the manuscript. D.F. fabricated samples. J.W., A.B., N.N. and D.P. performed the surface characterisation and participated in data analysis, D.F., D.B. and T.B. initiated the design of the

study and revised the manuscript. K.V. supervised the study and revised the manuscript.

## Appendix A. Supplementary data

Supplementary data to this article can be found online at https://doi.org/10.1016/j.mtbio.2021.100176.

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